

Contents lists available at ScienceDirect

Sensors and Actuators B: Chemical



journal homepage: www.elsevier.com/locate/snb

Selective C_2H_2 detection with high sensitivity using SnO_2 nanorod based gas sensors integrated with a gas chromatography



Jun Ho Lee^a, Min Sun Park^a, Hwaebong Jung^a, Yong-Sahm Choe^b, Wonkyung Kim^c, Young Geun Song^{d,e}, Chong-Yun Kang^{e,f}, Hyun-Sook Lee^{a,*}, Wooyoung Lee^{a,*}

^a Department of Materials Science and Engineering, Yonsei University, 50 Yonsei-ro, Seodaemun-gu, Seoul, 03722, Republic of Korea

^b iSenlab Inc., Halla Sigma Valley, Dunchon-daero 545, Jungwon-gu, Seongnam-si, Gyeonggi-do, 13215, Republic of Korea

^c School of Nano and Materials Science and Engineering, Kyungpook National University, 2559 Gyeongsang-daero, Gyeongsangbuk-do, 37224, Korea

^d Display and Nanosystem Laboratory, School of Electrical Engineering, Korea University, Seoul, 02841, Republic of Korea

^e Center for Electronic Materials, Korea Institute of Science and Technology (KIST), Seoul, 02791, Republic of Korea

^f KU-KIST Graduate School of Converging Science and Technology, Korea University, Seoul, 02841, Republic of Korea

ARTICLE INFO

Keywords: Acetylene gas sensor SnO₂ nanorod array Metal catalyst Miniaturized gas chromatography Dissolved gas analysis Transformer oil

ABSTRACT

We report on the sensitive and selective sensing properties of SnO_2 nanorods (NRs) based gas sensors coupled with a miniature gas chromatography (mini-GC) system for the detection of acetylene. The SnO_2 NRs were fabricated by a glancing angle deposition (GLAD) method and their average height and diameter were ~ 200 nm and ~ 30 nm, respectively. In order to overcome a selectivity issue of metal oxide semiconductor gas sensors, we integrated our SnO_2 NRs based sensors with a packed column. The device accurately and selectively detected acetylene within 2 min (~ 120 s). We found that loading a thin layer (5 nm) of metal catalysts such as Au, Pt or Pd increases the sensing abilities of the SnO_2 NRs sensors. Among the tested sensors, the Pd-coated SnO_2 NRs sensor (Pd-SnO₂ NRs) exhibited the best sensing performance for the detection of 10 ppm of acetylene and the lower detection limit of 0.01 ppm. The superior sensing properties of Pd-SnO₂ NRs are due to the large amount of oxygen deficiencies on the surface of Pd-SnO₂ NRs, which acts as reactive sites. More importantly, the mini-GC device can be used to selectively detect 10 ppm of acetylene from other gases such as H₂. Our findings demonstrate that Pd-SnO₂ NRs integrated with a mini-GC device can be utilized to monitor the dissolved acetylene gas in transformer oil in real time.

1. Introduction

Transformers are used worldwide to lower voltages and distribute electricity to individual houses. In order to decrease the internal temperature of transformers and to insulate currents, oils are being used in transformers. Since transformers have a long lifetime, there is a high risk for oils in older devices to degrade and damage the transformer [1–3]. This is because defects such as fractures, or discharges inside the transformer can cause the oil or cellulose to decompose and release various gases inside the transformer. These gases may include hydrogen (H₂), hydrocarbons (CH₄, C₂H₂, C₂H₄, C₂H₆) and carbon oxides (CO, CO₂) [1]. Of these gases, acetylene, C₂H₂, holds some serious potential risks. An accumulation of only small amounts of C₂H₂ (5–7 ppm) may cause arcing inside the transformer, which can lead to extremely dangerous situations such as explosions [2]. Hence, monitoring the concentration of C₂H₂ during operation is a critical issue for the operation of transformers.

Various devices are currently being used to examine fault gases inside transformer oils [3–7]. These methods can be divided into two groups: off-line and on-line techniques. In an off-line analysis, transformer oils are extracted from sampling valves of transformers and transferred to separate laboratories for analysis. To analyze the gas, dissolved gas analysis (DGA) or gas chromatography (GC) are usually utilized [3–5]. However, because of the time gap between the extraction and the analysis, the dissolved gases can diffuse to the atmosphere, leading to inaccurate results. On the contrary, using an on-line method, membranes are directly attached to transformer oils to separate gases from them [3,8]. This system is, however, quite expensive and complex to install in all transformers. Therefore, it is necessary to develop an economical device, which can perform real-time monitoring of gases dissolved in transformer oils with high sensitivity and selectivity.

In order to develop a sensitive device, it is crucial to find an efficient

* Corresponding authors.

E-mail addresses: h-slee@yonsei.ac.kr (H.-S. Lee), wooyoung@yonsei.ac.kr (W. Lee).

https://doi.org/10.1016/j.snb.2019.127598

Received 9 August 2019; Received in revised form 13 December 2019; Accepted 17 December 2019 Available online 23 December 2019 0925-4005/ © 2019 Elsevier B.V. All rights reserved.



Fig. 1. An illustration of the fabrication of (a) the bare SnO₂ NRs using a glancing angle deposition (GLAD) method via e-beam evaporation and (b) the resulting metal-coated SnO₂ NRs.



Fig. 2. Scheme of the concept of the selective gas sensing process of fault gases in transformer oils using the mini-GC system with an integrated Pd-coated $SnO_2 NR$ sensor. The fault gases are separated from the transformer oil by membranes and then injected into the mini-GC, after which the sensor selectively detects each gas component.

sensing material for C₂H₂. Extensive researches have been carried out to fabricate gas sensors based on metal oxide semiconductors (MOS) for the detection of C₂H₂ [9–19]. Especially, SnO₂ has been widely used as a sensor material for detecting various kinds of gases due to its wide bandgap (\sim 3.7 eV), high chemical and physical stability, and good sensitivity [10–14,18–23]. In order to improve the sensing performance, an increase in the specific surface area of the sensing material is generally employed by construing a nanostructure. There are many reports on the preparation of various types of nanostructures in SnO₂ [22]. In addition, decoration of noble metals, such as Au, Pt, and Pd,

could be added to further enhance the sensing properties because the metals serve as sensitizers or promoters. Most of the study presented that the metal-loaded SnO_2 exhibited a higher sensing response than pure SnO_2 .

In this context, we have recently demonstrated that Pd-coated SnO_2 nanorod arrays (NRs) synthesized by glancing angle deposition (GLAD) via e-beam evaporation possess notable sensing properties towards H₂ [24]. The Pd-coated SnO_2 NRs exhibited a superior sensing response compared to other SnO_2 based sensors with a detection limit of 0.2 ppm at room temperature. With this excellent performance as a gas sensor,



Fig. 3. SEM images of the (a) top and (b) cross-sectional views of Pd-coated SnO₂ NRs and elemental color mapping images for (c) Sn and (d) Pd using energy dispersive X-ray spectroscopy (EDS).

we decided to investigate its sensing characteristics to C_2H_2 , because MOS based gas sensors usually have high responses for various types of gases. Furthermore, since production of H_2 is accompanied by that of C_2H_2 , it is necessary for a H_2 gas sensor to overcome a selectivity issue, a common obstacle for MOS sensors. Introducing a column for GC to a gas sensing device would help overcome this problem by separating two gases.

In this work, we investigated the sensing properties of the bare and noble metal (Au, Pt and Pd) coated $SnO_2 NRs$ to C_2H_2 . We found that a noble metal on the surface of $SnO_2 NRs$ significantly enhanced their sensing properties. This was owing to the spill-over effect of the noble metals by creating a higher proportion of oxygen defect sites. We demonstrated that the Pd-coated $SnO_2 NRs$ sensor integrated with a GC column can sensitively and selectively detect acetylene in a mixed gas within 2 min.

2. Materials and methods

2.1. Synthesis and characterization of SnO₂ nanorod arrays

SnO₂ nanorod arrays (SnO₂ NRs) were fabricated on Al₂O₃ substrate (0.5 × 0.25 mm²) supplied with Pt interdigited electrodes and heating components by glancing angle deposition (GLAD) via e-beam evaporation (Fig. 1(a)). Commercial SnO₂ granules (4 N purity, Kojundo Chemical Laboratory Co., Ltd.) were used as SnO₂ source inside a 7cc-crucible and all conditions for the experiment were mentioned in a previous report [24,25]. The incident angle for the deposition was 80° with respect to the vapor flux and the substrate was rotating at a speed of 15 rpm. The growth rate was maintained at 1 Å/s and the process was initiated at an pressure of 5.0×10^{-6} torr. The process was conducted at room temperature and sensors were heat treated at 550 °C for 2 h afterwards.

Thin metal films (Au, Pt, Pd) were deposited on the top of the SnO₂ NRs using an ultra-high vacuum DC sputtering system (SNTEK Co., Ltd; Fig. 1(b)). The deposition was conducted at a pressure of 3.4×10^{-7} torr and 34 sccm of Ar was injected for initiation. The deposition rates of Au, Pt, and Pd were 0.67 nm/s, 0.28 nm/s, and 0.31 nm/s, at working powers of 30 W, 20 W, and 30 W, respectively.

The microstructures and the crystallinity of the fabricated SnO_2 NRs were investigated by field emission-scanning electron microscopy (FE-SEM, JSM-7100 F) equipped with an energy-dispersive X-ray spectroscope (EDS). Surface elemental analysis was conducted by X-ray photoelectron spectroscopy (XPS, K-alpha Thermo U. K.) using Al K α radiation.

2.2. Sensor characterization

The mini-GC system previous reported in detail [26,27] was used for all gas sensing experiments. In brief, it was equipped with a sensor based on SnO_2 NR, a packed gas chromatography column (GC column), a sampling loop, a mini-sized pump and three solenoid valves. The actual dimensions of the device were $8 \times 13 \times 16$ cm³ and the length and the inner radius of the column were 90 cm and 0.1 cm, respectively. The column was filled with Porapak Q (Restek), and both ends of fillers were sealed by piles of glass wool.

The selective detection procedure of the mini-GC device integrated with the SnO₂ NR based sensor is described in Fig. 2. We operated the mini-GC device with an integrated SnO₂ NR based sensor by first injecting 1 mL of target gas (C_2H_2) mixed with interfering gases in ambient air at room temperature into the sampling loop by mini-pump without pre-concentration. Consequently, the mixed gas was divided into each component after passing the packed column because the strength of interaction between each gas molecule and the column filler was different. The gases with non-polarity (e.g. H_2) and smaller size are



Fig. 4. (a) Sensor response (Δ sensor signal) of the device integrated with bare SnO₂ NRs and the Au-, Pt-, and Pd-coated SnO₂ NRs (5 nm) for an acetylene concentration of 10 ppm as a function of operating temperatures. (b) Variation of the sensor signal as a function of time for different Pd thicknesses (3, 5, and 7 nm). The inset presents the sensor response of Pd-coated SnO₂ NRs prepared with different Pd thickness at 200 °C.

released from the column faster than others because of the weak polarity of the stationary phase coated on the column filler. Thereafter, the target gas was separately detected within 3 min by the SnO₂ NRs based sensors. The operating temperature of sensors were varied for the different sensing materials based on their optimal working temperature, but that of the column was set to 25 °C.

In the mini-GC system, response of a sensor was defined as follows [26,27]:

Δ SensorSignal = log(Resistance)_{max} - log(Resistance)_{min}

where *log(Resistance)max* and *log(Resistance)min* represent the resistance of the sensor when exposed to ambient air and the target gas, respectively. These values were derived from the peak height in the gas chromatograms corresponding to the relative response of the sensing material.

3. Results and discussion

In order to investigate the microstructure and the elemental distribution of the as-synthesized catalyst-coated SnO_2 NRs, we carried out FE-SEM and EDS analysis. Fig. 3(a) and (b) shows the SEM images of

the top and cross-sectional views of the Pd-coated SnO₂ NRs fabricated on Al₂O₃ substrate, respectively. The nanorods were randomly distributed on the entire surface in clusters and the distance between the clusters was about 5–20 nm (Fig. 3(a)). Since the size of a single grain inside the cluster was about 30 nm in diameter, we concluded that the average diameter of a single nanorod was ~ 30 nm. The nanorods were vertically attached to the surface of the substrate and the average height of the nanorods was ~ 200 nm as depicted in the cross-sectional view of the Pd-coated SnO₂ NRs (Fig. 3(b)). EDS elemental color mapping images on the cross-sectional view of the sample (Fig. 3(c) and (d)) confirmed that the Pd nanoparticles were evenly deposited on the surface of the SnO₂ NRs.

In order to optimize the sensing performance of the SnO₂ NRs toward acetylene, we first examined the optimal working temperature and thickness of the metal catalysts of the as-synthesized SnO₂ NRs. We determined the optimal working temperature by measuring the peak heights (Δ Sensor Signal) of the bare and the different catalyst-coated SnO₂ NRs upon exposure to 10 ppm of C₂H₂ at increasing temperatures (Fig. 4(a)). The optimal working temperatures for the bare and the Au-, Pt-, and Pd-coated SnO2 NRs (5 nm) were 350 °C, 400 °C, 200 °C, and 200 °C, respectively. Since SnO₂ NRs sensors can degrade at a temperature above 350 °C in long-term use, however, the optimal temperature for the Au-coated SnO2 NRs was set to 350 °C, instead of 400 °C. To optimize the thickness of the catalyst coating, we used the Pdcoated SnO₂ NRs because they showed the highest sensing response in the previous test. By investigating coating thickness of 3, 5, and 7 nm, we found that depositing 5 nm of Pd led to the best sensing performance (Fig. 4(b)). We used the same thickness for the other metal coatings as well.

To evaluate the acetylene sensing performances of the as-synthesized bare and Au-, Pt-, and Pd-coated SnO_2 NRs integrated in the mini-GC system, we analyzed the changes in the sensor signal over time. The sensor signals of the samples were measured at their respective optimal working temperature at different C₂H₂ concentrations in the range of 0.01–50 ppm in dry air (Fig. 5). All SnO₂ NRs showed single peaks indicating the presence of C₂H₂ at around 120 s. Among the tested sensors, the bare SnO₂ NRs exhibited the lowest sensing properties and the metal-coating improved their responses.

To evaluate the sensing abilities of the SnO₂ NR sensors to acetylene, we estimated the sensing response (Δ Sensor Signal) from the sensor signal curves at the various C_2H_2 concentrations. Fig. 6(a) shows Δ Sensor Signal against concentration curves (both in log scales) for all fabricated SnO₂ NRs. All curves exhibited linearity and this was mainly due to an exponential relationship between sensor response and concentration [28]. The Pd-coated SnO2 NRs, exhibited the best sensing performance at 10 ppm or lower concentrations of acetylene. At 10 ppm C_2H_2 , the sensing response was ~0.06, ~0.52, ~0.47, and ~0.99 for the bare SnO2 NRs and the Au-, Pt-, and Pd-coated SnO2 NRs, respectively (Fig. 6(b)). Similarly, the detection limit was the lowest for the Pd-coated SnO₂ NRs (0.01 ppm) compared to the bare (1 ppm) and the Au- (0.1 ppm), and Pt-coated SnO₂ NRs (0.05 ppm; Fig. 6(b)), respectively. Accordingly, we found that the metal-coated SnO₂ NRs had significantly improved sensing abilities compared to the bare SnO₂ NRs. Therefore, it could be inferred that metal nanoparticles improved overall properties of sensors dramatically. As a result, the Pd-coated SnO₂ NRs had the best sensing performance, such as highest sensing response at C₂H₂ concentration below 10 ppm, the lowest detection limit, and the lowest optimal working temperature compared to other samples. Detecting C₂H₂ at concentrations of less than 10 ppm is crucial to prevent malfunction inside a transformer. Therefore, the Pd-coated SnO₂ NRs seem to be the best option among our investigated samples for the use in a gas analyzer of dissolved C₂H₂ in transformer oil.



Fig. 5. Sensor signal of (a) bare, (b) Au-coated, (c) Pt-coated, and (d) Pd-coated SnO₂ NRs as a function of time for various C₂H₂ concentrations at their respective operating temperatures.

To analyze the chemical composition and oxidation states of elements in sensing materials, we performed XPS analysis (Fig. 7(a)–(d)). From XPS results, we found that all samples showed the peaks for Sn 3d (~500 eV), Sn 3p¹ (~760 eV), and O 1s (~530 eV) [29] confirming the successful fabrication of the SnO₂ NRs. Additionally, XPS peaks for Au (e.g. Au 4f (~100 eV), Au 4d⁵ (~340 eV), Au 4d³ (~360 eV) and others), Pt (e.g. Pt 4f, Pt 4d⁵, Pt 4d³ and others), and Pd (e.g. Pd 4p, Pd 3d, Pd 3p¹) were observed in the Au-, Pt-, and Pd-coated SnO₂ NRs, respectively. This indicates the presence of metal nanoparticles in each sample and as a consequence, successful deposition of each metal [30,31].

To investigate the cause for the differing sensing performances of the various metal coated NRs, we assessed the ratio of the oxygen vacancies in different samples. For this, we deconvoluted the O1s peaks in the XPS results of the bare, Au-, and Pt-coated SnO_2 NRs into three individual Gaussian peaks at about 530 eV (blue), 531.5 eV (green), and 532 eV (pink) (Fig. 7(e)–(g)). The peaks correspond to O^{2-} ions in SnO– bonds of SnO₂ structure, oxygen deficient sites (oxygen vacancies, and oxygen interstices), and oxygen atoms chemisorbed on the surface of SnO₂, respectively. For the Pd-coated SnO₂ NRs, we used six different peaks to deconvolute the peak at ~528 eV–540 eV because the Pd 3p peaks occurred in the same region [30] (Fig. 7(h)). The ratio of the peak assigned to oxygen vacancies varied from 19.31% in the bare SnO_2 NRs to, 25.29% in the Au-, 29.42% in the Pt-, and 32.28% in the Pd-coated SnO_2 NRs. The great increase of deficient oxygen in the metal-coated SnO_2 NRs compared to the bare NRs can be attributed to the spill-over effect of the noble metal [28]. Because this tendency correlates that of the sensing performance, the density of oxygen deficient sites such as oxygen vacancies could be related to the sensing performance of the NRs. This is corroborated by a previous study reporting that oxygen vacancies were responsible for the sensitivity of ntype MOS gas sensors [32].

Based on our findings, we propose the following sensing mechanism of Pd-coated SnO_2 NRs (illustrated in Fig. 8). In ambient air, a thicker electron depletion layer is formed inside them compared to the bare SnO_2 NRs (Fig. 8, left). This is because the work function of Pd metal is higher than that of the SnO_2 semiconductor, so that more electrons belonging to SnO_2 are transferred to Pd [33,34]. Therefore, the dissociation of the oxygen molecules into oxygen ion species at the surface of the Pd nanoparticles is enhanced. Consequently, due to the spill-over effect of Pd, the large number of oxygen ion species easily diffuses to the surface of the SnO_2 NRs [33,34], resulting in the more reactive sites for the detection of the target gas. Further, the thicker electron depletion layer limits the conduction paths of electrons inside the SnO_2 NRs, and consequently increases the resistance. As the sensor is exposed to a C_2H_2 environment, C_2H_2 molecules react with the oxygen ion species



Fig. 6. (a) The sensor response of the different SnO_2 NR based sensors as a function of C_2H_2 concentration in log scales. (b) Columnar diagrams of both Δ Sensor Signal (blue, left y-axis) and detection limit (orange, right y-axis) of all sensors (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article).

and produce CO_2 and H_2O (Fig. 8, right). During these reactions, electrons are released back to the conduction band of the SnO_2 NRs. This leads to a decrease in the depth of depletion layers, resulting in a decrement of resistance. Pd nanoparticles might also acted as active

sites for the activation of target gas molecules, thus catalyzing sensing reactions. Therefore, change of resistances for SnO_2 NRs based sensors in air and C_2H_2 environment made the detection possible.

Finally, in order to verify selectivity, we tested the selective C₂H₂ detection of Pd-coated SnO₂ NRs with a mixture gas of 10 ppm C₂H₂ and 100 ppm H₂ at an operating temperature of 200 °C. To compare the signals, air, 100 ppm H_2 , and 10 ppm C_2H_2 were also separately injected and evaluated (Fig. 9). H₂ was chosen as an interfering gas since it is another main default gas frequently evolved from transformer oil. As a result, when 100 ppm H_2 and 10 ppm C_2H_2 were singly injected, a single peak was observed at 20 s and 120 s, respectively. For the gas mixture of H₂ and C₂H₂, we could detect both peaks clearly separated indicating the high acetylene sensing selectivity of the Pd-coated SnO₂ NRs. This characteristic implies that the selectivity can be achieved from gas sensors based on metal oxide semiconductor by integrating with a GC column. We believe that this device would be applicable to an on-line sensing system for diffused gases from transformer oils filtered by membrane filters, thus enabling a real-time monitoring system for transformers.

4. Conclusion

In conclusion, we report on a miniaturized gas chromatography (GC) device equipped with a SnO2 NR-based sensor for accurate and selective detection of C₂H₂, a critical fault gas dissolved in transformer oils. We found that Au-, Pt-, and Pd-coated SnO2 NRs exhibited an improved sensing performance compared to the bare SnO2 NRs. In particular, Pd-coated SnO2 NRs with a coating thickness of 5 nm showed the best sensing response to 10 ppm C₂H₂ at the lowest optimal working temperature of 200 °C. Pd-coated SnO₂ NRs were able to detect 0.01 ppm C₂H₂ within 120 s and to selectively detect C₂H₂ in a mixture with H_2 (10 ppm $C_2H_2/100$ ppm H_2). The dramatically improved performance of the Pd-coated compared to the bare SnO₂ NRs was mainly attributed to an increase of oxygen deficient sites on the surface of the metal-coated SnO₂ NRs owing to the spill-over effect of the noble metal nanoparticles. In conclusion, we successfully demonstrate that by integrating Pd-coated SnO₂ NRs into a GC column, the selectivity issue of acetylene sensors can be overcome. We anticipate that the device can be utilized for real-time monitoring of evolved gases from transformer oils.



Fig. 7. Full XPS spectra of (a) bare SnO₂ NRs, (b) Au-SnO₂ NRs, (c) Pt-SnO₂ NRs, and (d) Pd-SnO₂ NRs. (e–h) Deconvolution of the O 1s peak at 525 eV–540 eV for each sensor (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article).

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Fig. 9. The sensor signal of the mini-GC device integrated with Pd-SnO₂ NRs when exposed to air, 100 ppm H_2 , 10 ppm C_2H_2 , and a gas mixture of 100 ppm H_2 and 10 ppm C_2H_2 at 200 °C.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgements

This research was supported by the Medium and Large Complex Technology Commercialization Project through the Commercialization Promotion Agency for R&D Outcomes (2019K000045) funded by the Ministry of Science, ICT & Future Planning, and the Priority Research Centers Program through the National Research Foundation of Korea (NRF) (2019R1A6A1A11055660). Fig. 8. Sensing mechanism of Pd-coated SnO₂ NRs. In ambient air condition (left), electrons react with oxygen molecules to form oxygen species on the surface the of SnO2 NRs as well as thick electron depletion layers. Pd nanoparticles (pink) catalyze the reaction owing to a spill-over effect. When exposed to C₂H₂ (right), the C₂H₂ molecules react with the oxygen species and produce CO2, H2O, and electrons. Electrons move back to SnO2 NRs and decrease electron depletion layers. The resistance of Pd-coated SnO2 NRs decreases during the process (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article).

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Jun Ho Lee received a Bachelor's degree and a Master's degree in Material Science and Engineering at Seoul National University in 2015 and 2017, respectively. Since 2017, he is working on development of miniaturized GC integrated with metal oxide based gas sensors as a researcher at Yonsei University.

Min Sun Park received a Bachelor's degree in Chemistry at Sungkyunkwan University in 2017. Since 2018, she is working on development of the breath analyzer and miniaturized GC integrated with metal oxide based gas sensors as her Master's degree at Yonsei University.

Hwaebong Jung received a Bachelor's degree in Material Science and Engineering at Yonsei University in 2012. He is currently studying on the breath analyzer using metal oxide based gas sensor as a step toward his Ph.D degree at Yonsei University.

Yong-Sahm Choe is a CEO of iSenLab Inc. specialized for R&DB of breath analyzers. He received his B.S, M.S. and Ph. D degrees from the department of metallurgical engineering of Yonsei University in 1989, 1991, 1999, respectively and worked at TYM R&D institute from 1999 to 2011 in Korea. He developed oral malodor diagnosing devices, TWIN BREASOR and TWIN BREASOR II. His research interests are a development of breath analyzers utilizing semiconductor gas sensors and a miniaturization of gas chromatography for breath analyzers.

Wonkyung Kim was born in 1976 in Mokpo, Republic of Korea. He received a Ph. D. degree in Material Science and Engineering at Yonsei University in 2018. He has been a adjunct professor of School of Nano & Materials Science and Engineering at Kyungpook National University. In recent years, his research interests have centered on gas sensors.

Young Geun Song received his B.S. form Pusan National University, Korea, in 2016. He is currently a Ph.D. course student in the Department of Electrical Engineering, Korea University. His research interests include fabrication of metal oxide nanostructure, and chemical sensor applications.

Chong-Yun Kang received his Ph.D. from the Department of Electrical Engineering of Yonsei University in 2000. Now he is a Principal Research Scientist in KIST from 2000 and a professor of KU-KIST Graduate School of Converging Science and Technology in Korea University from 2012. His research interests include smart materials and devices, expecially, piezoelectric energy harvesting and actuators, electrocaloric effect materials, and nanostructured oxide semiconductor gas sensors.

Hyun-sook Lee received her Ph.D. from the Department of Physics of POSTECH in 2008. Now she is a research professor in the Department of Materials Science and Engineering of Yonsei University from 2015. Her research interests are various materials related in hightemperature superconductivity, solid-state hydrogen storage, rare-earth/rare-earth free permanent magnets, nanostructured metal oxide semiconductor gas sensors, and nanostructured thermoelectrics.

Wooyoung Lee is the Dean of School of Materials Science and Engineering and the Director of Institute of Nanoscience and Nanotechnology at Yonsei University in Korea. He received a BS degree in Metallurgical Engineering from the Yonsei University in 1986, a MS degree in Physics from University of Cambridge, United Kingdom in 2000. He is a regular member of National Academy of Engineering of Korea. He was a member of National Science & Technology Council and a director in Korea Israel Industrial R&D Foundation. In recent years, his research interests have centered on hydrogen sensors, various metal oxide semiconducting gas sensors, and breath analyzers. He is also studying thermo-electric materials and devices, and permanent magnets. He has received a number of awards in nano-related research areas and a Service Merit Medal (2008) from the Government of Korea due to contribution on the development of intellectual properties. He has authored and co-authored over 200 publications, and has edited three special books on nano-structured materials and devices.